

# **Imaging Composite Materials using X-Rays**

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Columbia University and IBM Research Division

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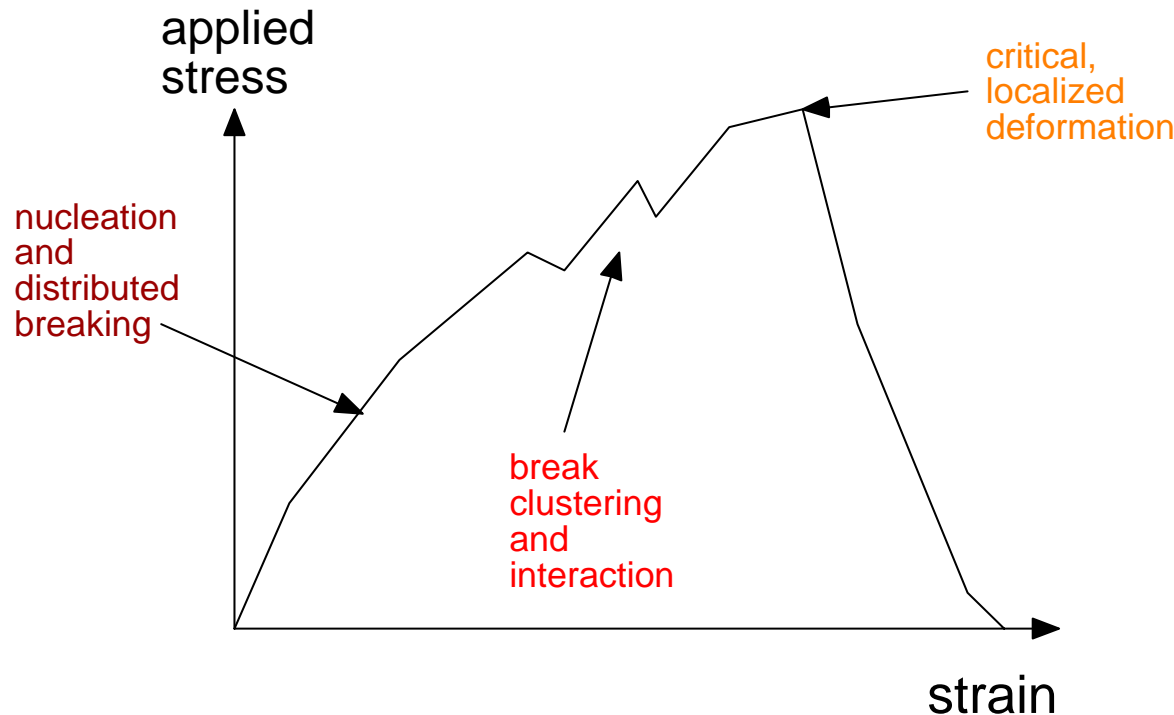
# Outline

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- Introduction
- Metal matrix composites (Ti-SiC and Al-Al<sub>2</sub>O<sub>3</sub>)
- Ceramic matrix composites (SiC-SiC)
- Metallic glass matrix composites
- Cracks in ferroelectrics
- *Study of damage evolution in complex materials*



# Fracture of a Fiber Composite under Tension



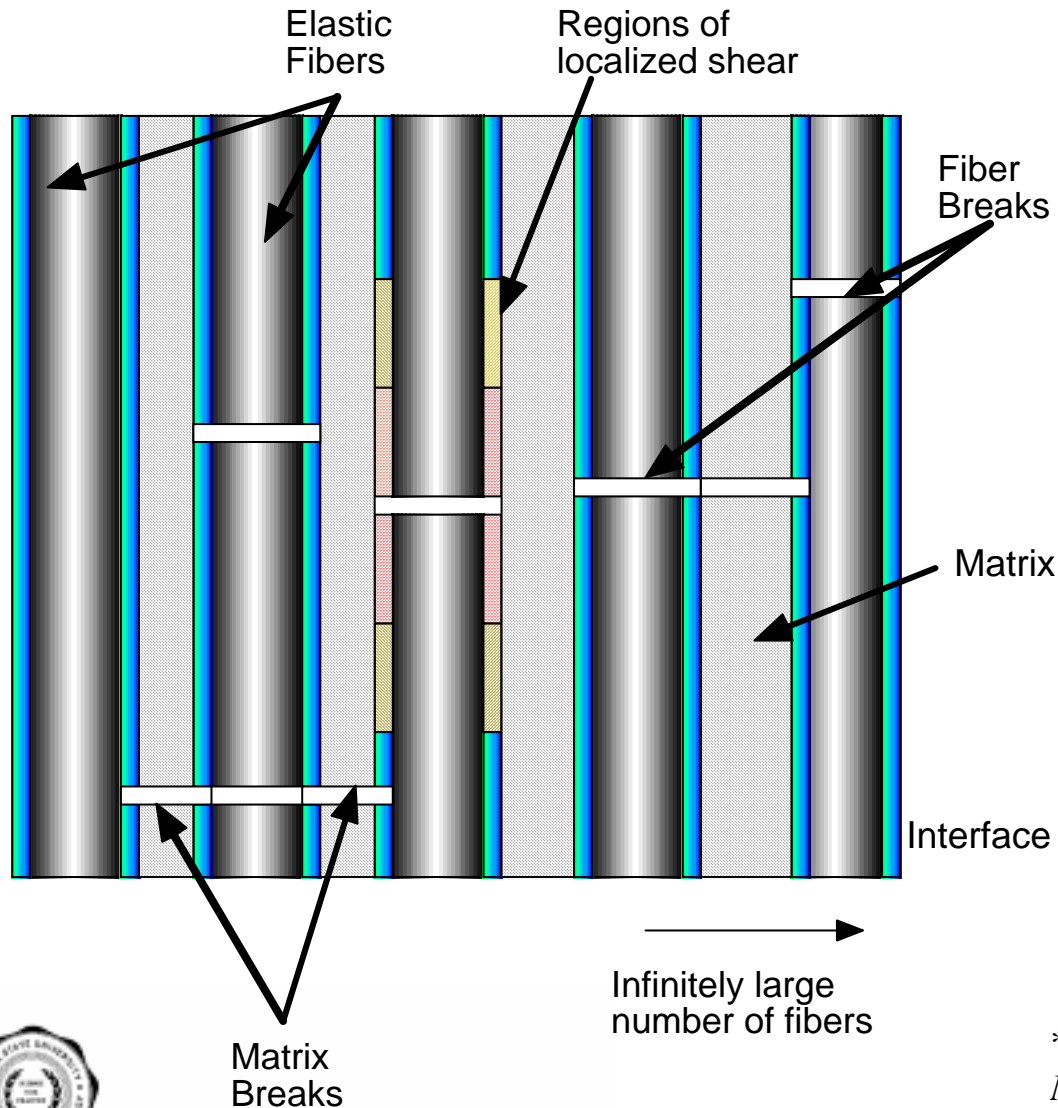
## Complications

- Fabrication processes
- Inhomogeneous dislocation densities
- Changes in grain size
- Geometrical constraints
- Interface introduced with different properties
- Residual stresses

- **Aim:** prediction of strength and lifetime
- **Need:** “realistic” constitutive laws and quantitative damage characterization



# Deformation of a Multi-Fiber Composite



## Critical Parameters:

- Location and morphology of cracks
- Crack opening displacement
- Size of plastic zone
- Debond size at the interface
- Strain concentrations near cracks and debonds

\* I.J. Beyerlein and C. M. Landis,  
*Mechanics of Materials*, 1999; 31: 331.



# Motivation and Approach

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- ❑ Little information about deformation and **constitutive behavior of materials at multiple length scales**.
- ❑ Need to **link experimental data with rigorous micromechanics modeling**.
- ❑ **Approach**: Use **X-ray diffraction and imaging** to investigate deformation in materials and complement it with **modeling**.
- ❑ **Critical issues**:
  - Need for model specimens
  - “High selectivity” of diffraction
  - Only elastic lattice strains are measured with diffraction
  - Lack of “realistic” constitutive laws to calculate stress and interpret diffraction data



# Advantages of XRD

- Non-destructive.
- Ability to distinguish different phases.
- Can measure elastic strain and texture.
- Simultaneous strain and imaging capability.
- Multi-scale: *nm* to *cm*.
- Deep penetration.
- *In-situ* experiment capability.

⇒ Determination of *in-situ* constitutive behavior

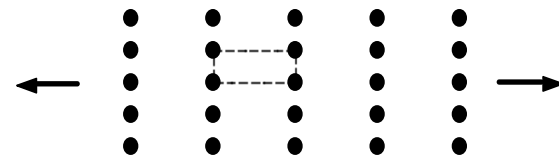
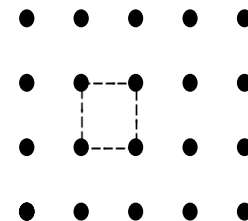
Bragg's law:

$$\lambda = 2d \sin \theta$$

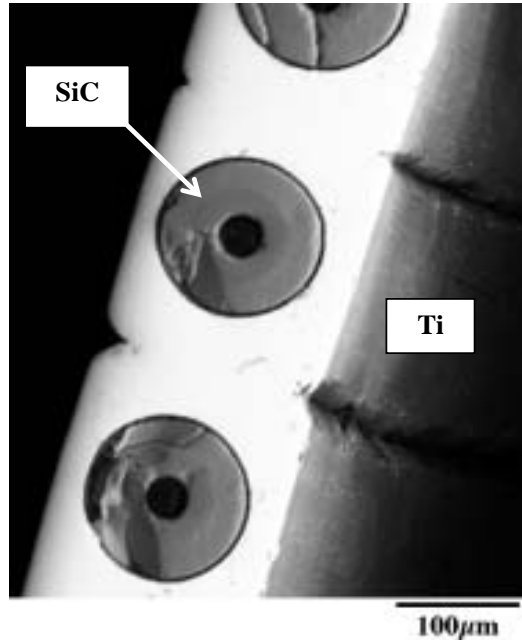
Differences in lattice spacing

⇒ Elastic lattice strain

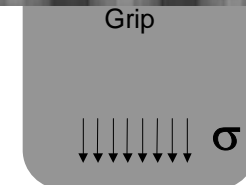
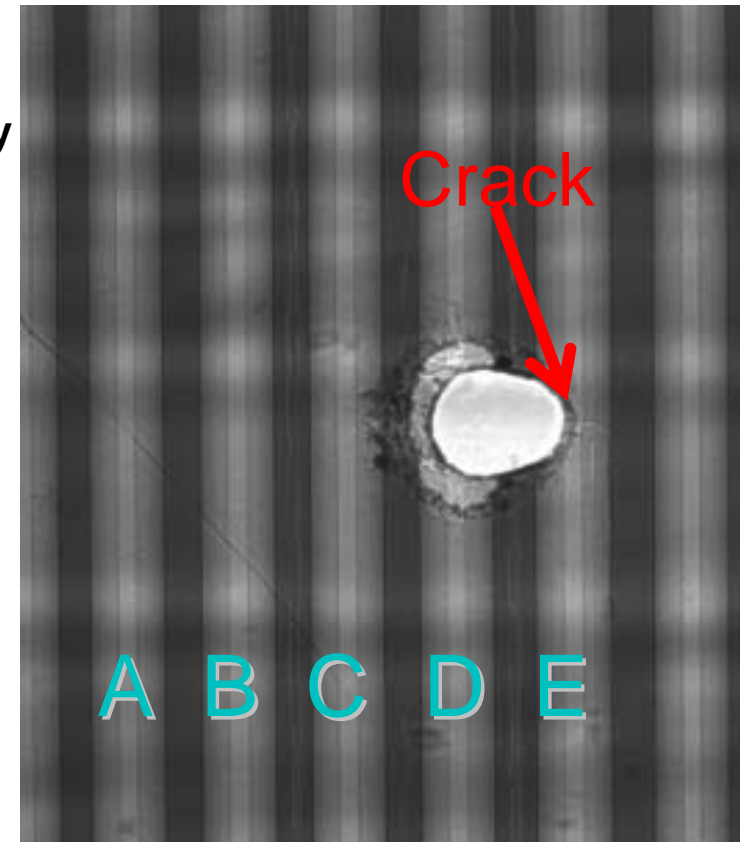
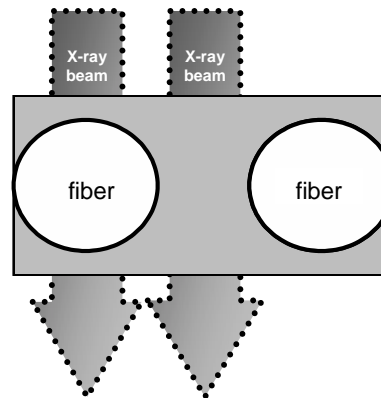
$$\varepsilon_{hkl}^{el} = \frac{d_{hkl} - d_{hkl}^0}{d_{hkl}^0} = \frac{d_{hkl}}{d_{hkl}^0} - 1$$



# Model Composite: Ti-6Al-4V / SiC (SCS-6)



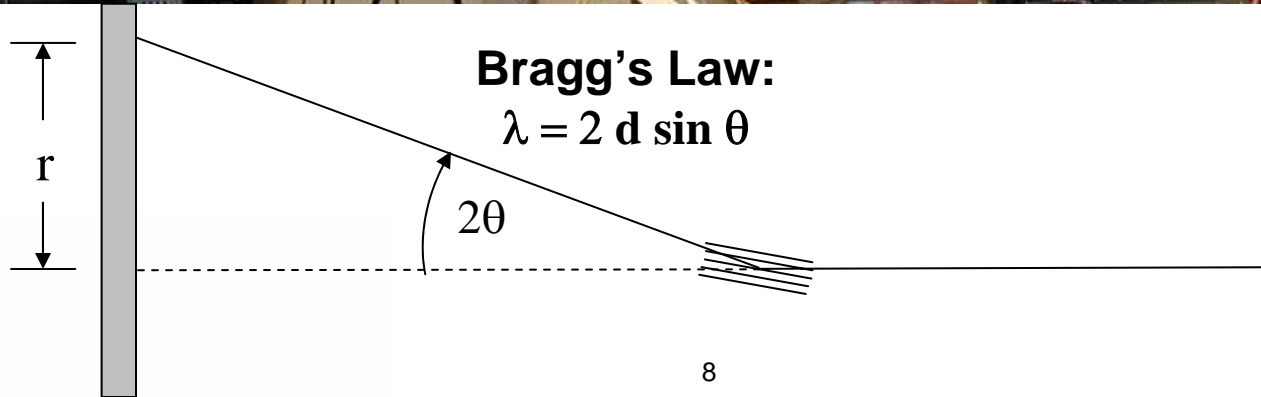
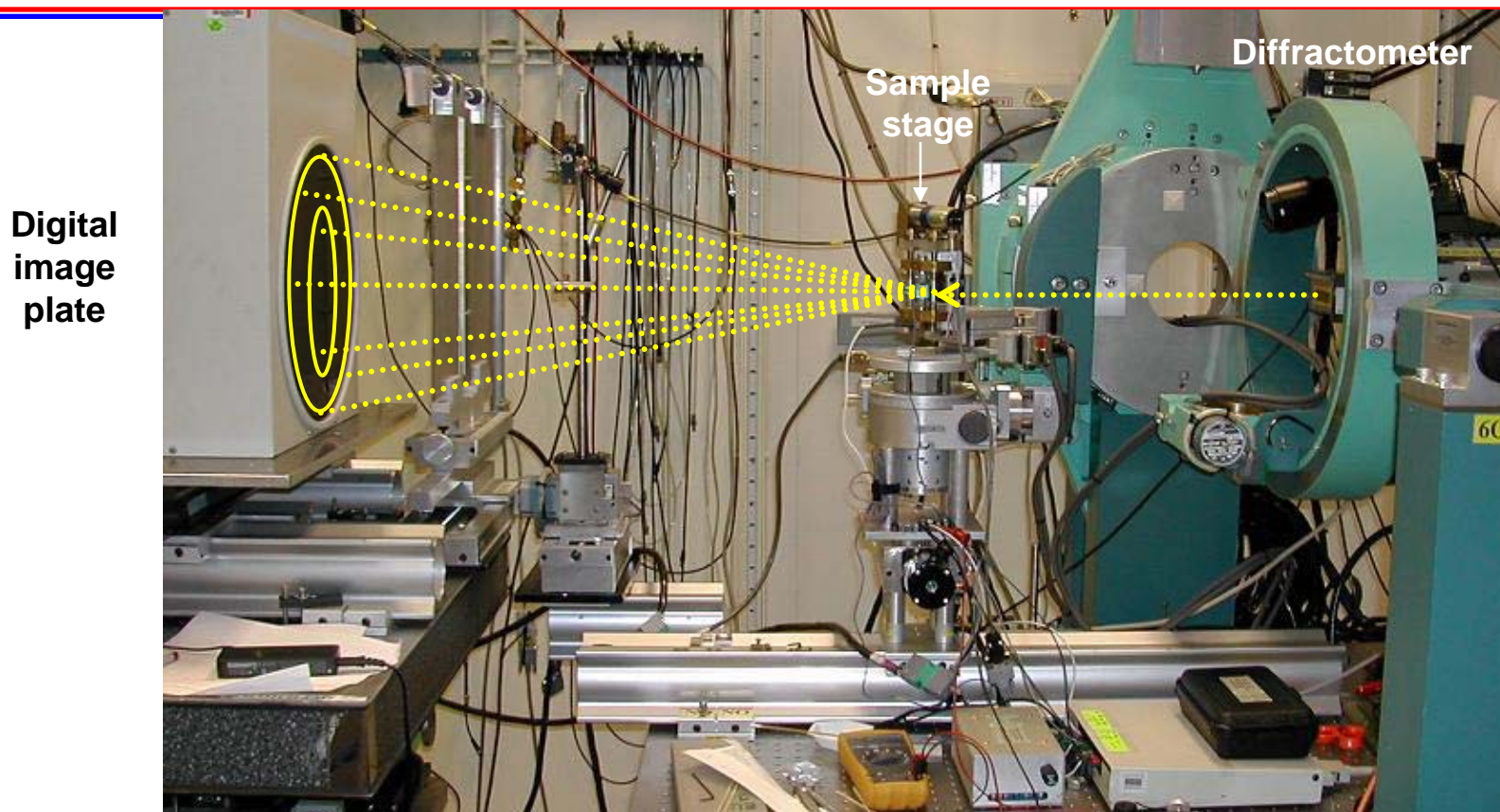
- ❑ Uniaxial tensile testing
- ❑ Damage evolution study using XRD (65 keV)
- ❑ Complete penetration
- ❑ 90 x 90 μm<sup>2</sup> spot size



- Laminar composite: Ideal for model comparison
- 140 μm in diameter fibers; 240 μm average center-to-center distance
- 200 μm thick matrix
- Data collected with a digital image plate



# High Energy 2-D XRD Experimental Setup

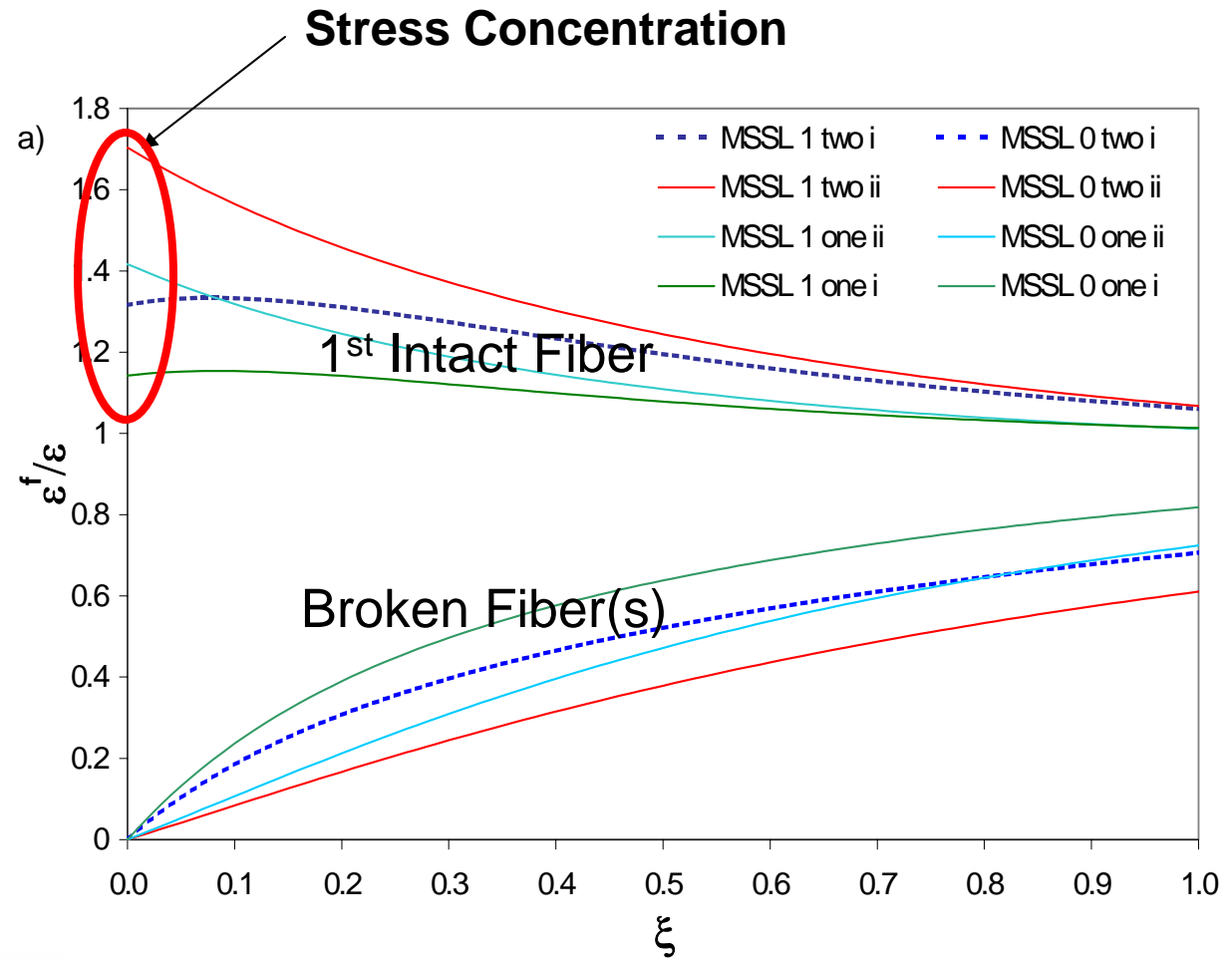




# Matrix Stiffness Shear Lag (MSSL) Model Predictions

- (i)-**intact** matrix
- (ii)-**broken** matrix

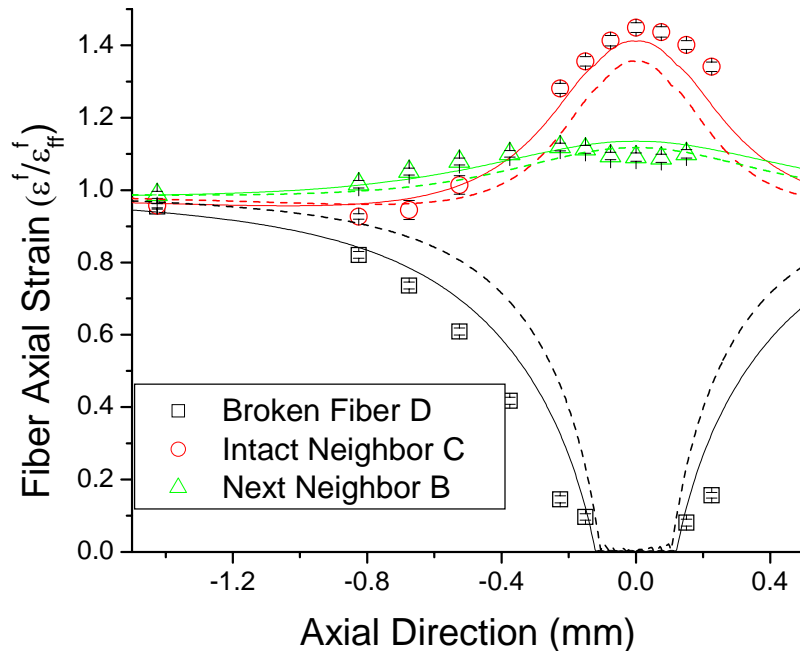
With  $\rho = 0.289$   
for both one  
and two broken  
fibers



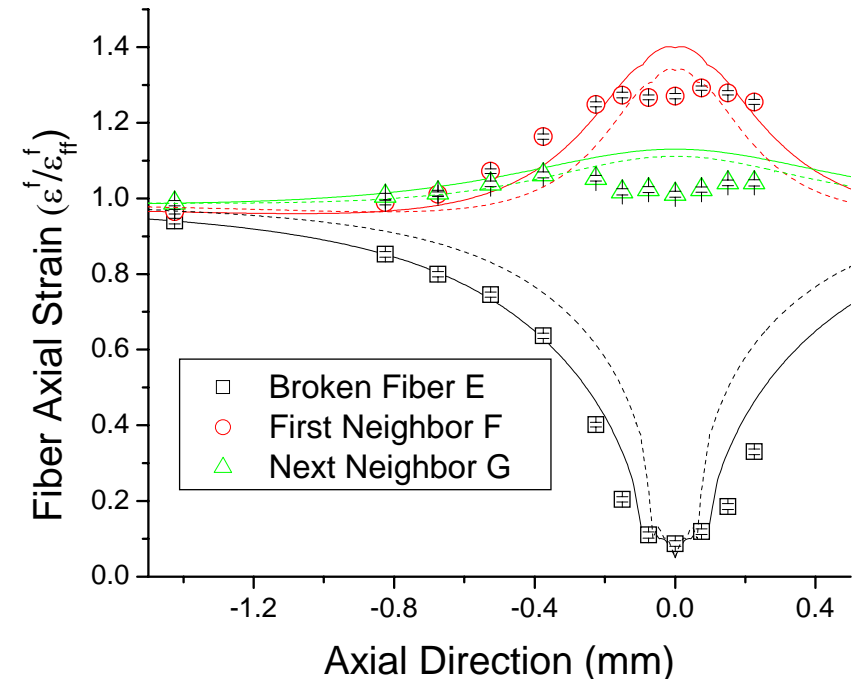
# Unloading Strains in Fibers Compared to the MSSL Model

Ti-SiC

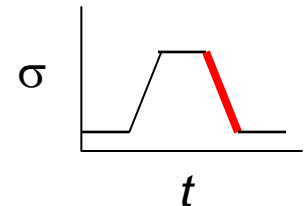
Left Side of the Damage Zone:



Right Side of the Damage Zone:



- Good fit with 'intact matrix' case.
- Unloading strains were used due to plasticity in matrix.
- Right hand side data suggest interface debonding.



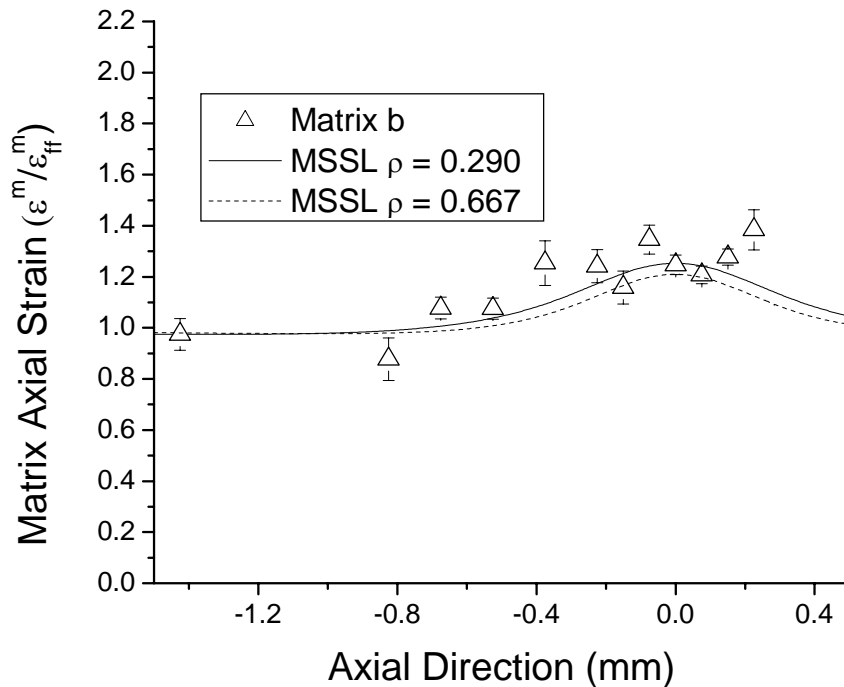
J. C. Hanan, E. Üstündag *et al*, *Acta Mater.* 51 [14], 4237-4248 (2003).



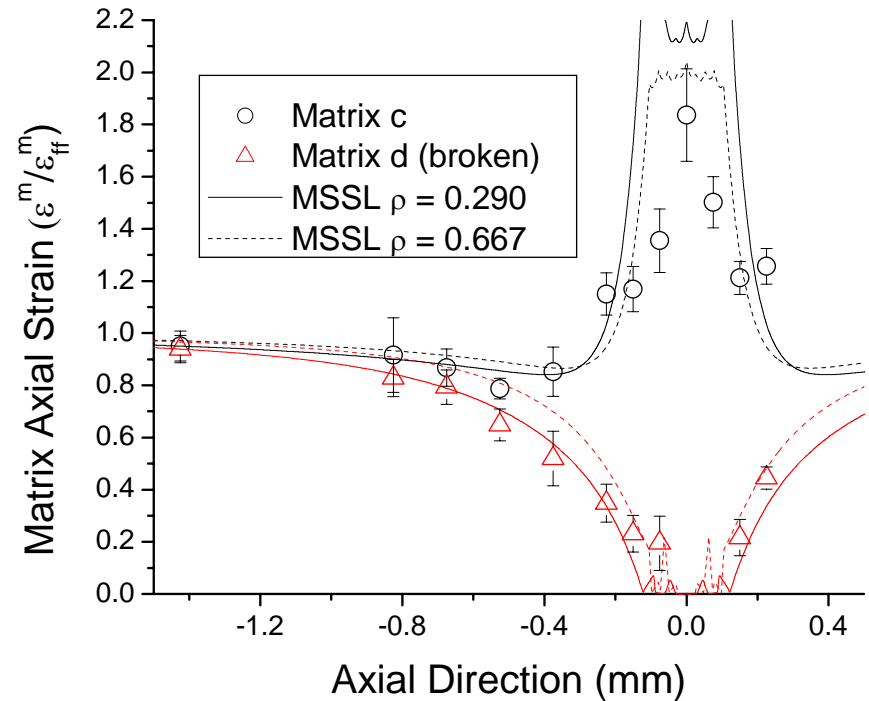
# Unloading Strains in the Matrix Compared to the MSSL Model

Ti-SiC

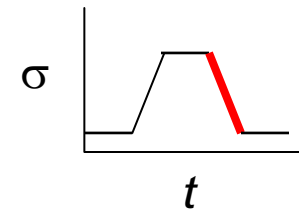
Matrix between two intact fibers:



Matrix around damage zone:



- Better fit with 'intact matrix' case.
- $\rho = 0.290$  appears to be a more realistic value.
- Matrix data comes from few grains.



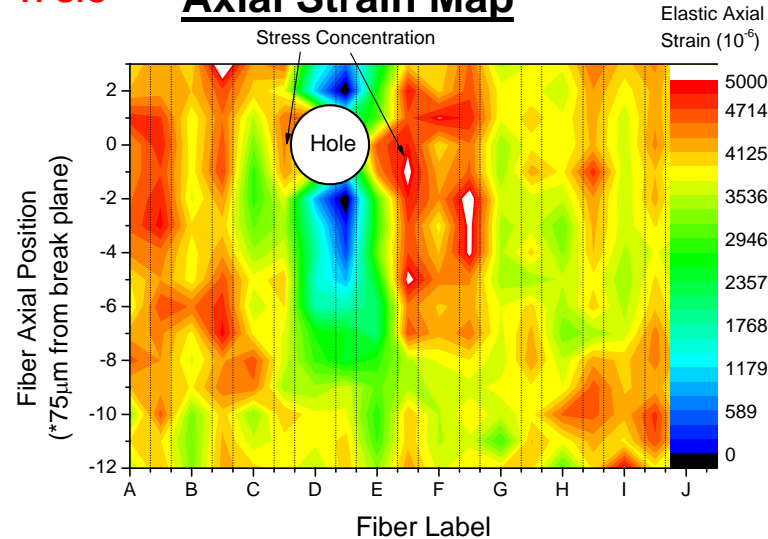
J. C. Hanan, E. Üstündag *et al*, *Acta Mater.* 51 [14], 4237-4248 (2003).



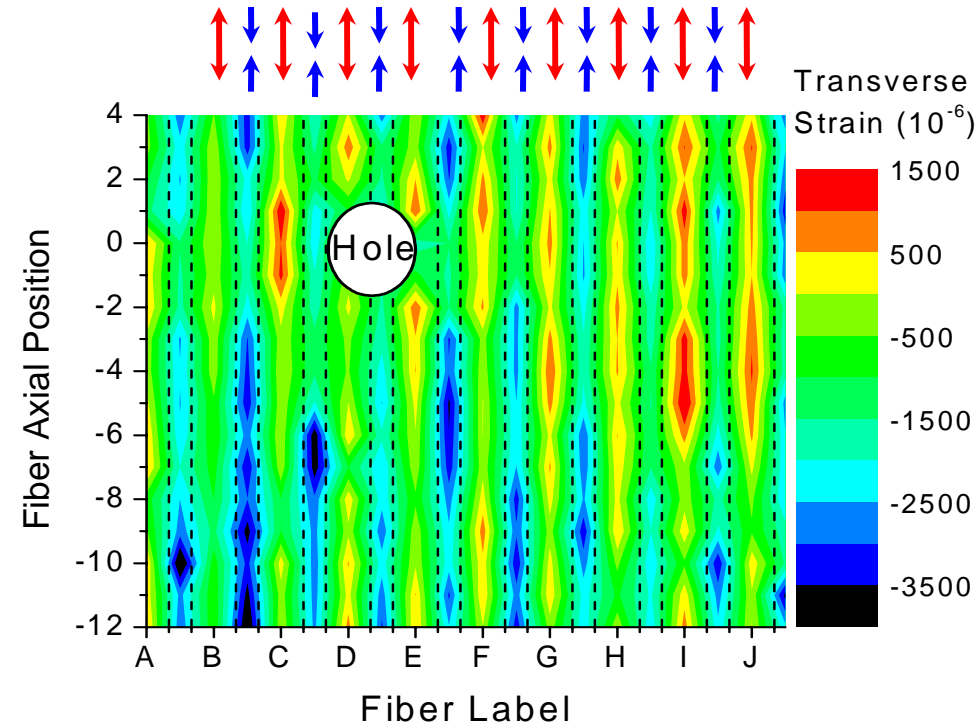
# Matrix Strains using Image Plate

Ti-SiC

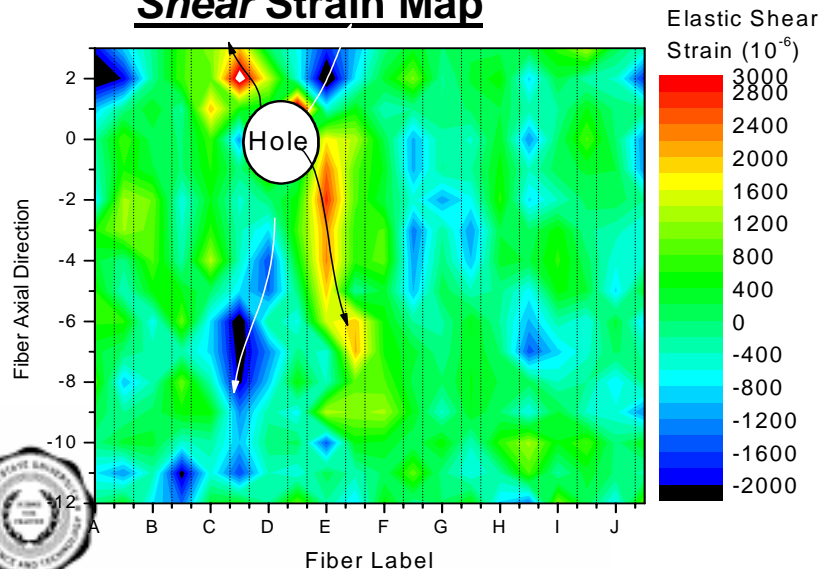
## Axial Strain Map



## Transverse Strain Map

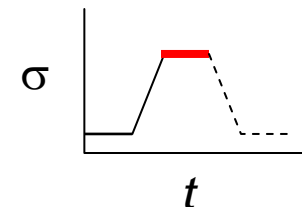


## Shear Strain Map



- Multi-axis strain data
- Significant strain concentrations in matrix

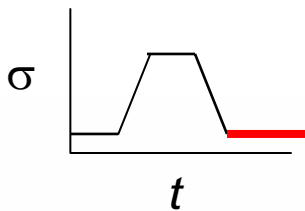
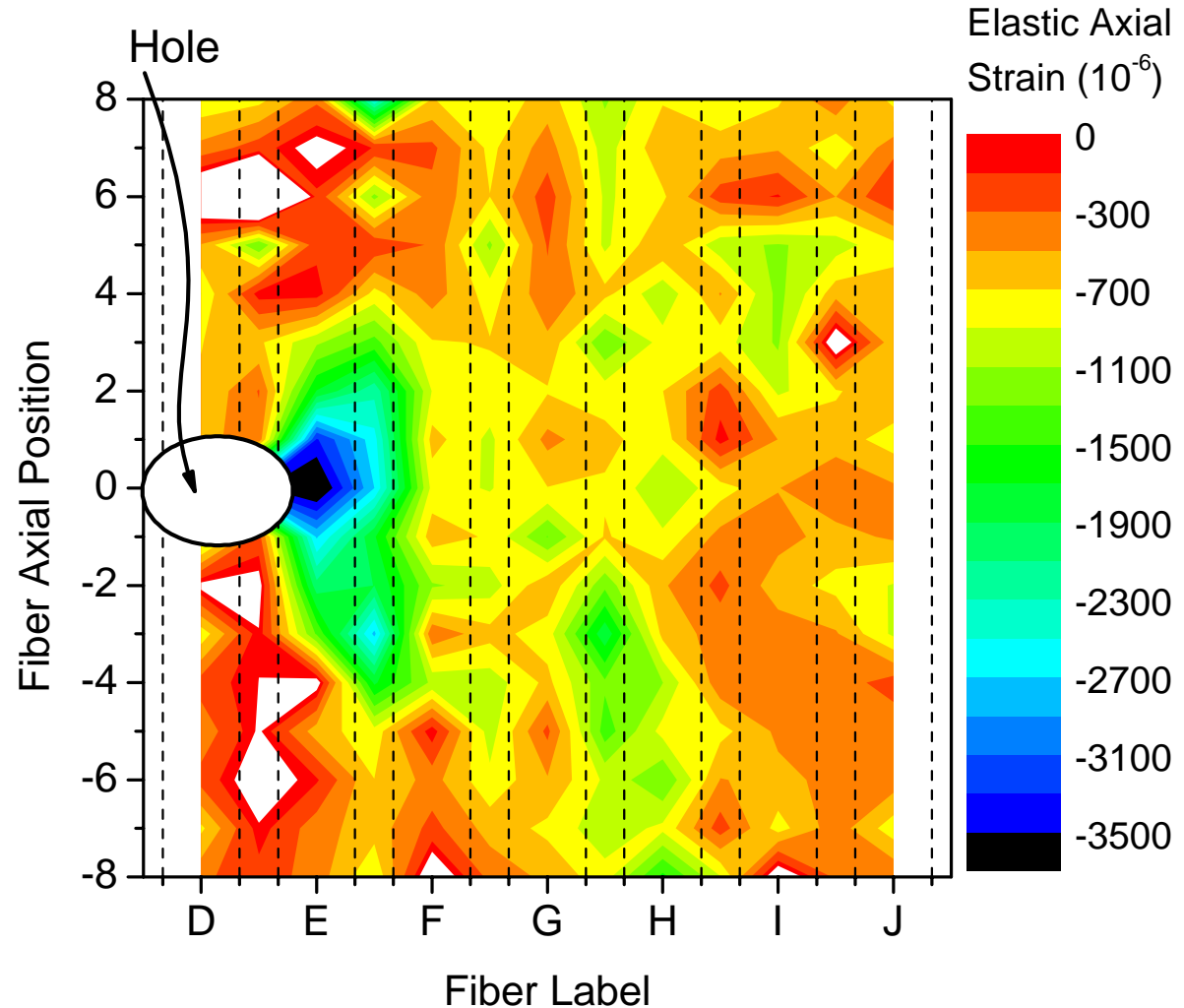
$$\sigma = 850 \text{ MPa}$$



# Change in Matrix *Axial* Residual Strain due to Loading

Ti-SiC

- The compressive regions identify plastic deformation while loading the composite

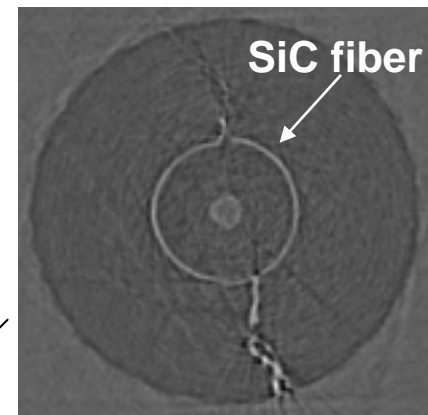
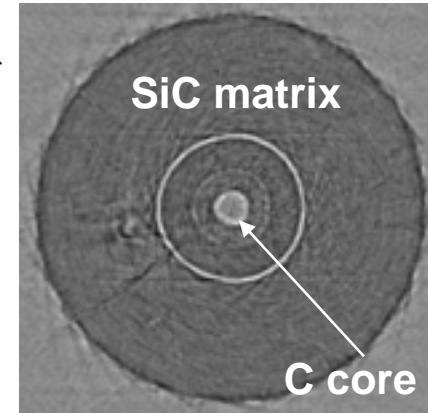


# SiC/SiC Composite: *Microtomography*

- Use of highly coherent X-ray beam (at APS).
- Contrast due to absorption and phase changes.
- Resolution  $\sim 1\ \mu\text{m}$ .
- Ideal for damage evolution studies in CMCs:
  - » Multi-dimensional strain information
  - » Identification of cracks/debonds



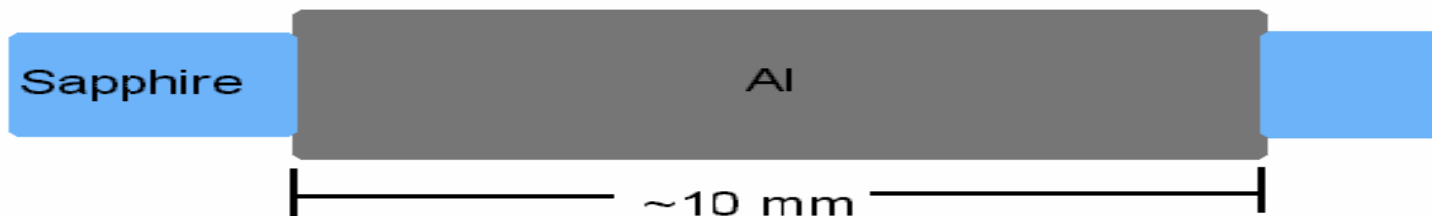
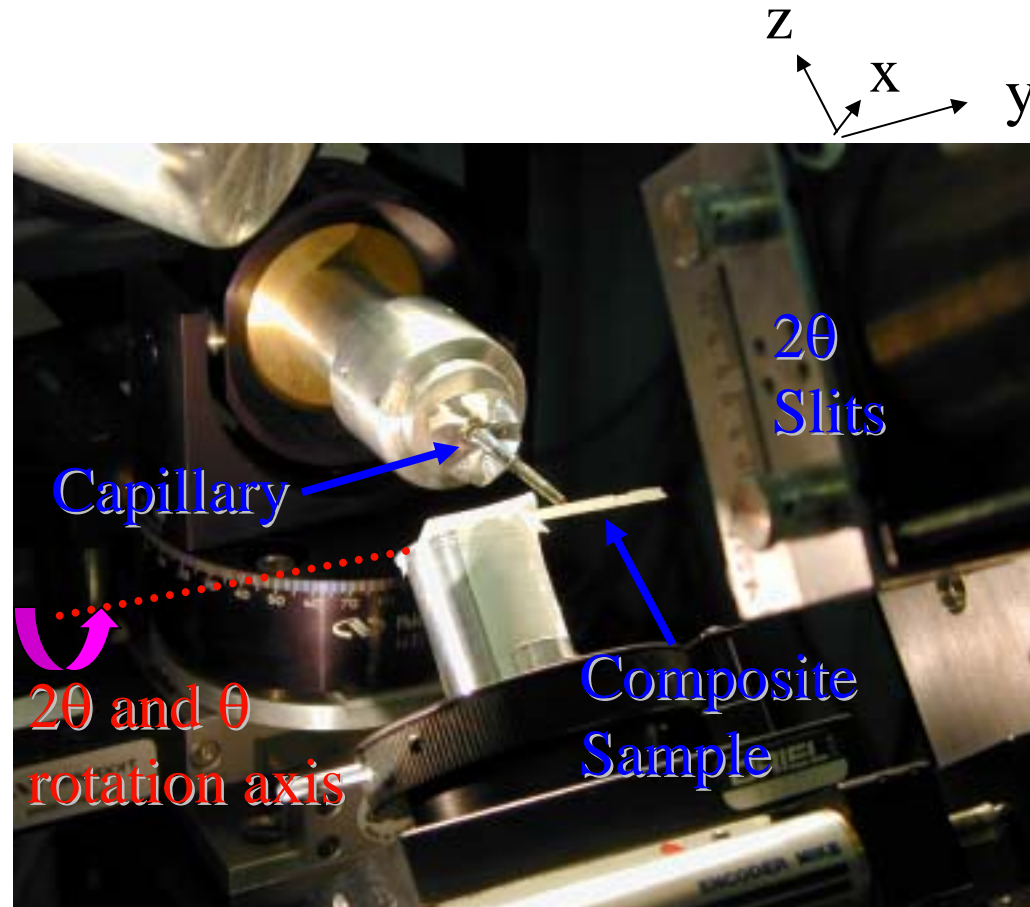
**Special load frame for tomography**



**Future: Combined diffraction and tomography**

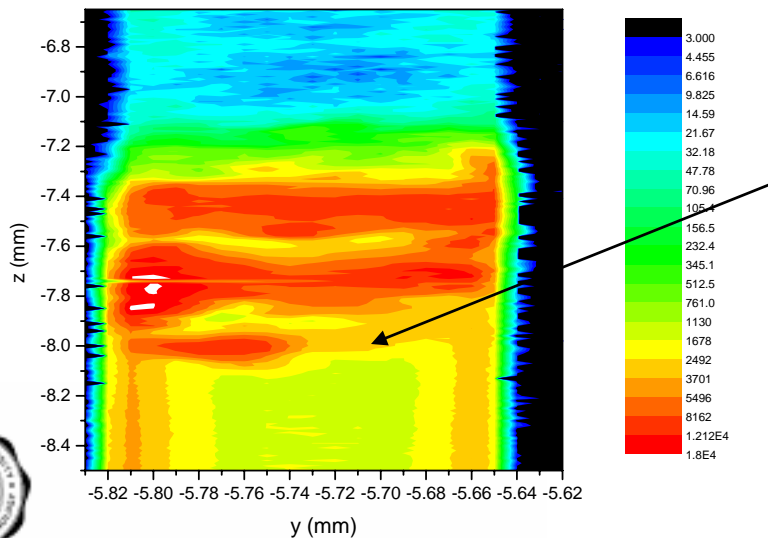
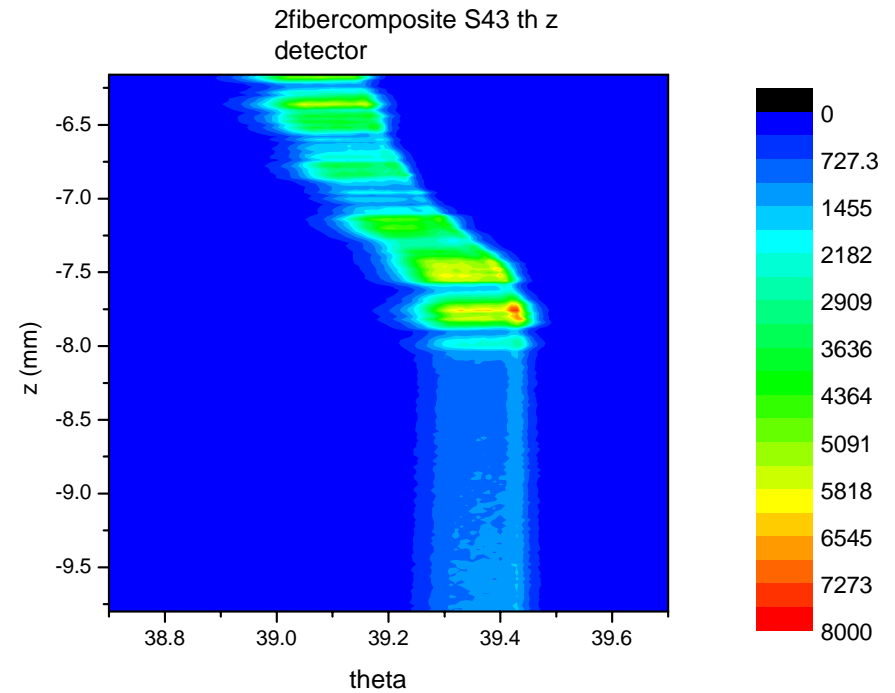
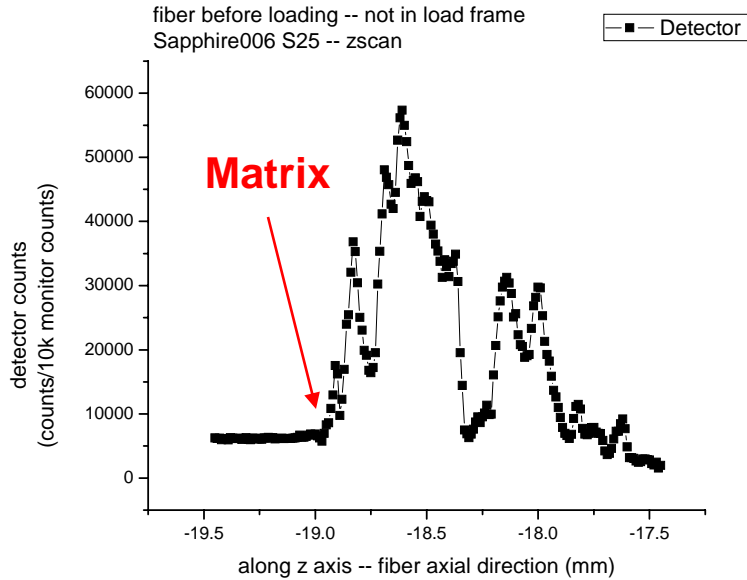
# X-Ray Microtopography

- National Synchrotron Light Source (BNL).
- Simultaneous diffraction, fluorescence and absorption.
- Microtopography provides information on the internal structure of composite:
  - matrix uniformity
  - interface integrity
  - fiber structure
- Spot size 2-10  $\mu\text{m}$  using capillary.
- Trajectory of the X-ray beam





# Effect of Matrix

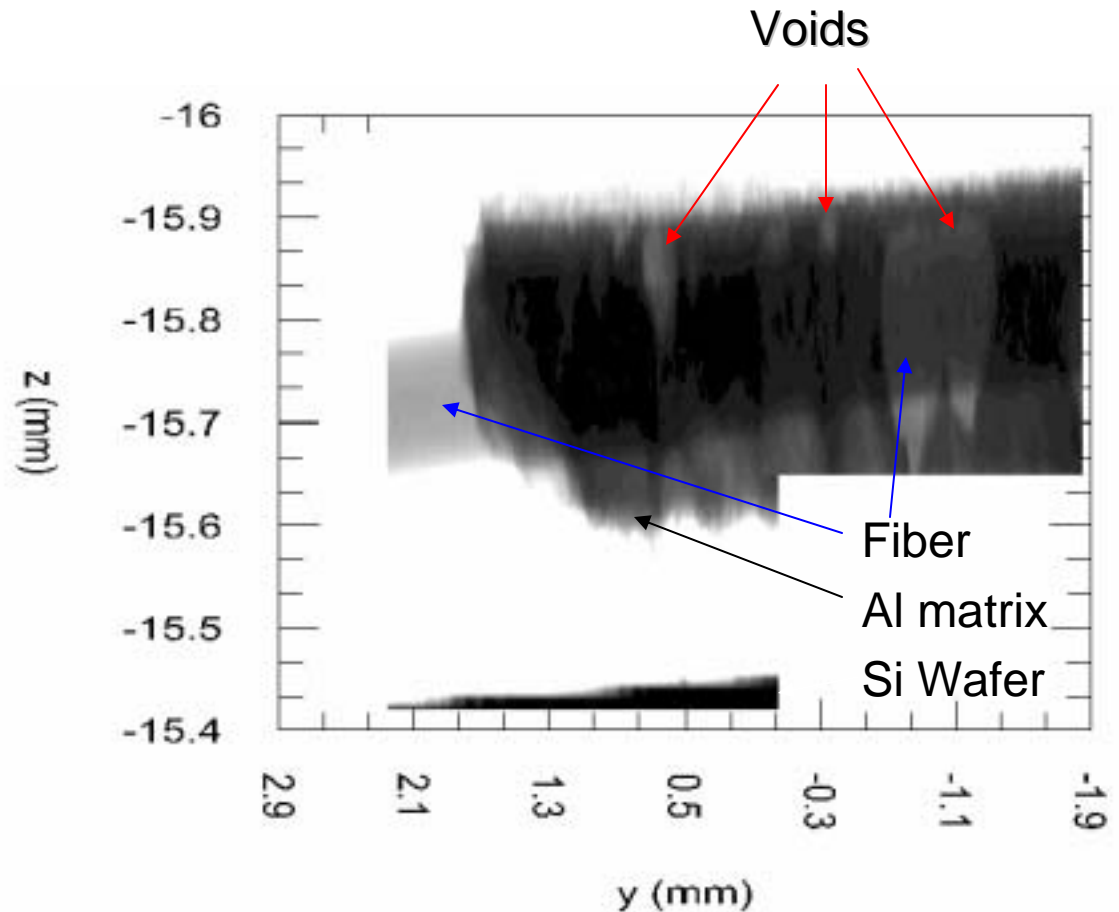


“Bending” of fiber inside  
matrix



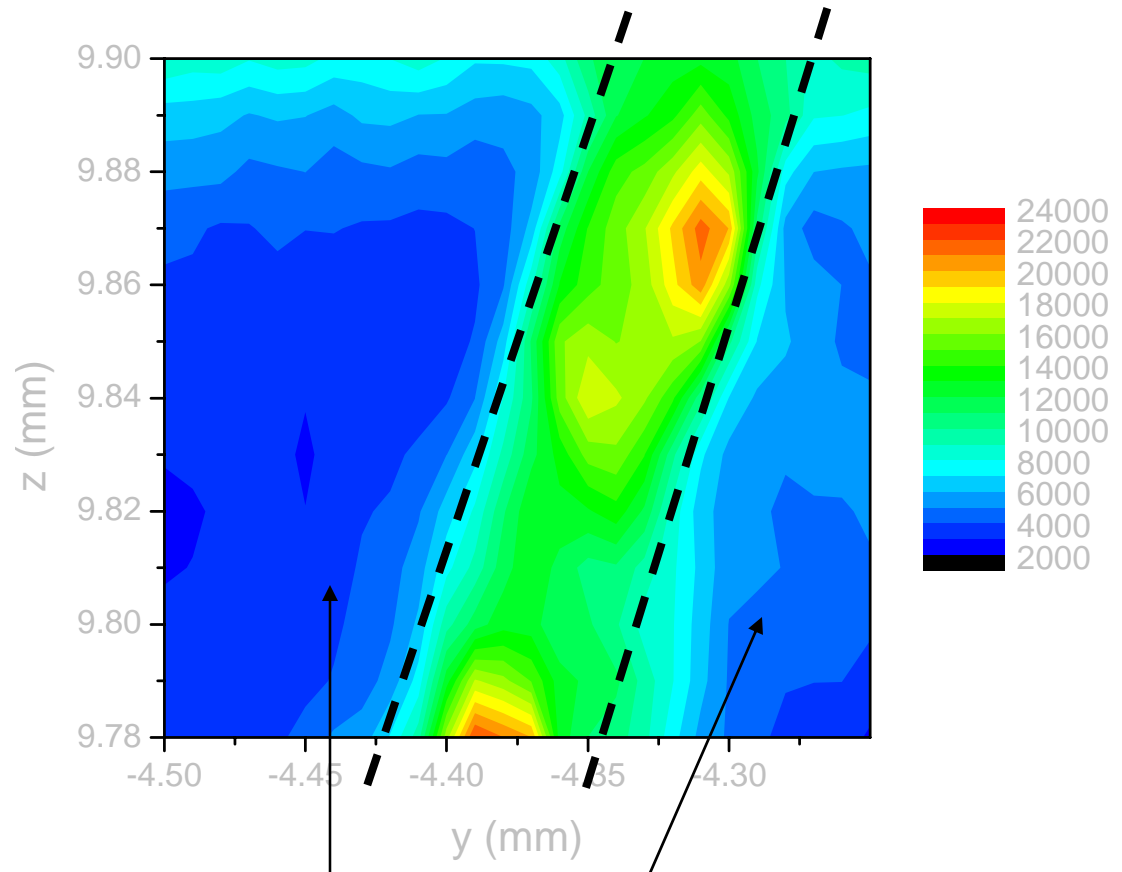
# Transmission Profile

- $\theta = 0^\circ$ ,  $2\theta = 0^\circ$ .
- Some small voids are visible. The largest is located above the legend in the graph.
- The sample was fixed to the wafer on the opposite end of the sample from the one shown.



# Radiograph of a Fiber Break

- Transmission reveals a  $75\text{ }\mu\text{m}$  gap between the fibers.
- An *in-situ* measurement of a crack opening in a fractured fiber.



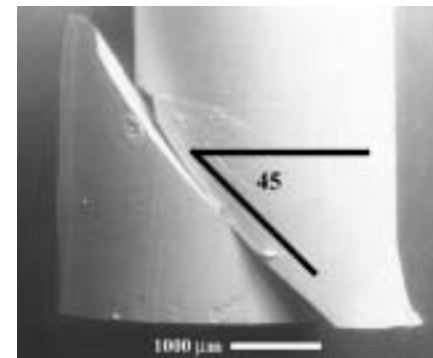
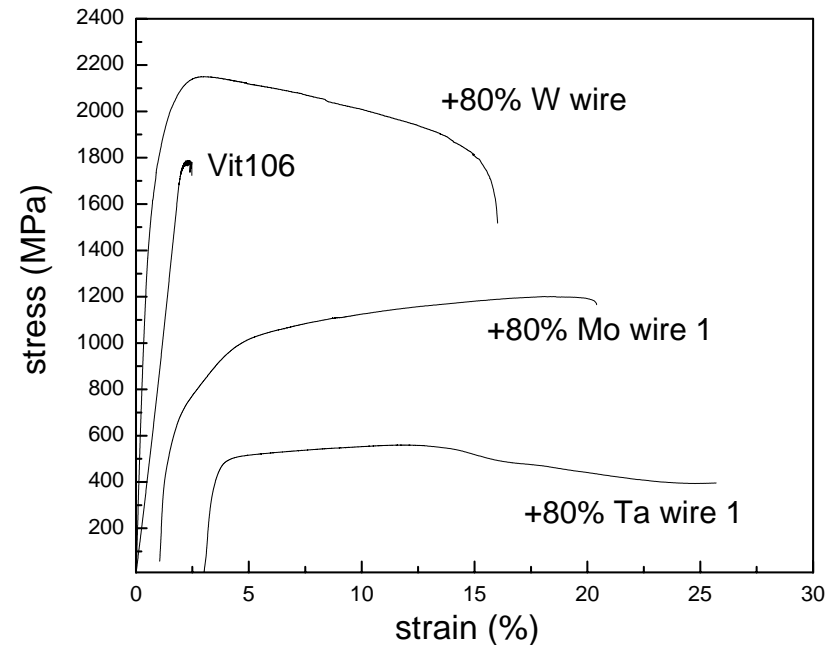
Broken fiber ends



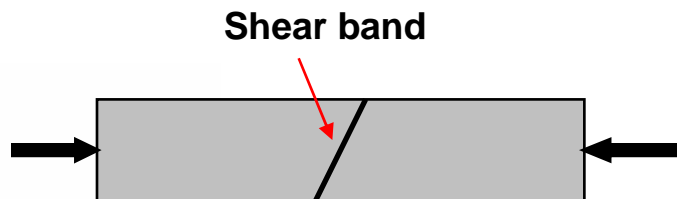
# Bulk Metallic Glass Matrix Composites

- ❑ Main problem with BMGs: **catastrophic failure** under unconstrained loading.
- ❑ Main **deformation mechanism** is via **shear bands** (at room T).
- ❑ Addition of reinforcements has been shown to increase damage tolerance and toughness.
- ❑ Critical questions:
  - » What is the *in-situ* mechanical behavior of reinforcements?
  - » How do reinforcements interact with shear bands?

## Compressive loading of fiber-reinforced BMG composites



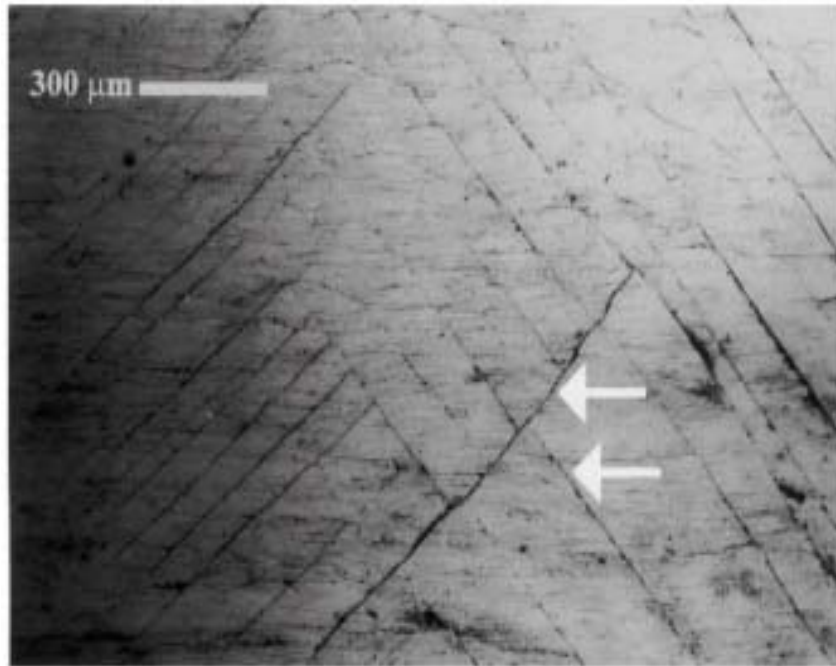
Monolithic  
BMG



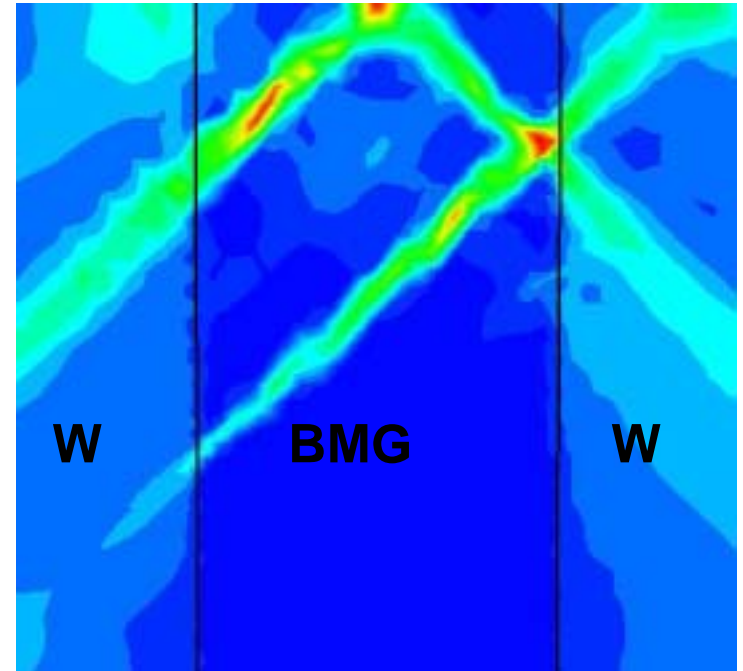
# W-Fiber / BMG-Matrix Composites:

## Damage Evolution

Multiple shear bands in 40% W-BMG composite

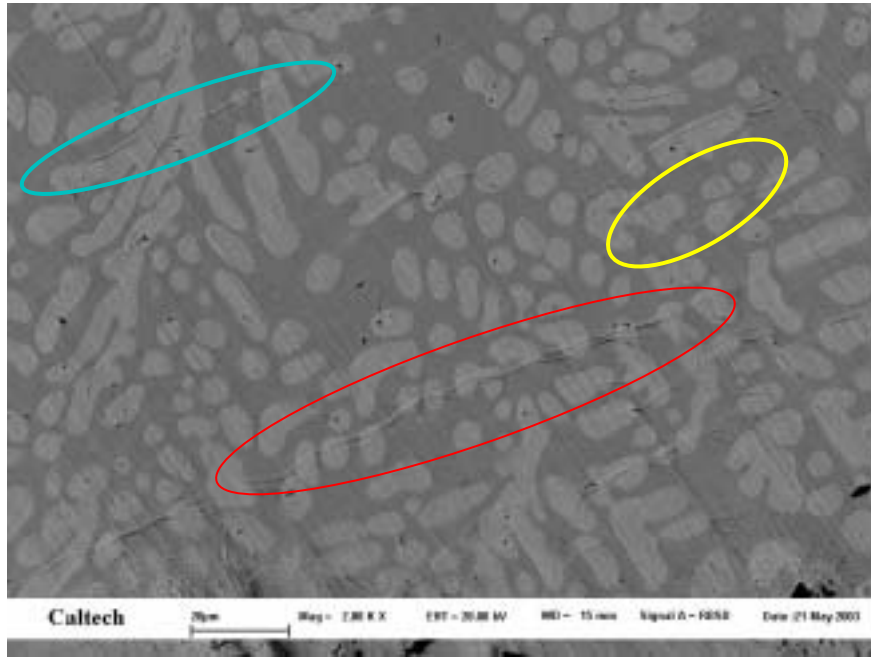


Modeling shear bands in W-BMG composites



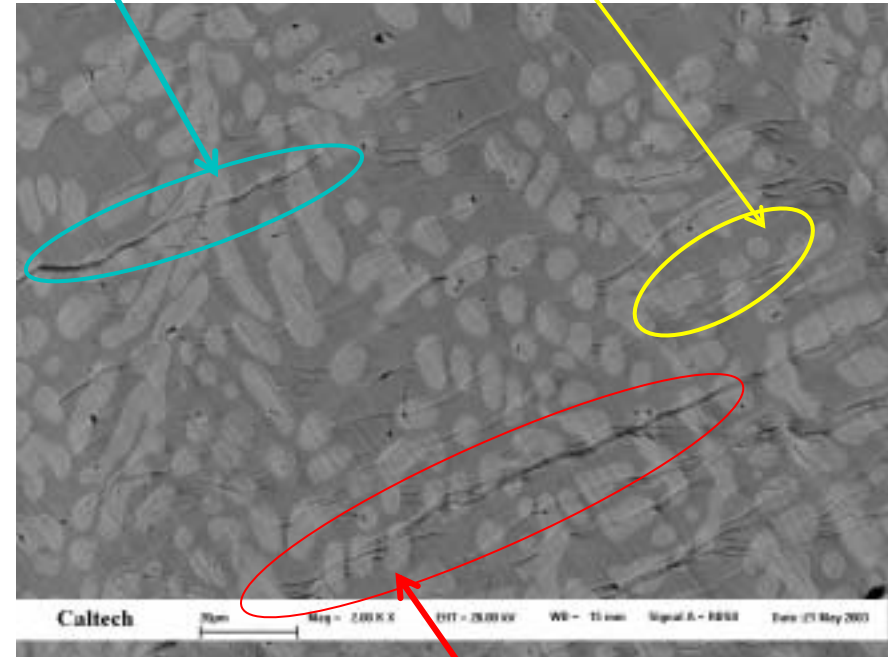
- BMG “yields” by multiple shear banding
- Slip initiates in W, transferred to BMG across interface
- *Details of micromechanics?*

# “ $\beta$ Phase” / BMG Composites: SEM Investigations



Connect

Nucleate



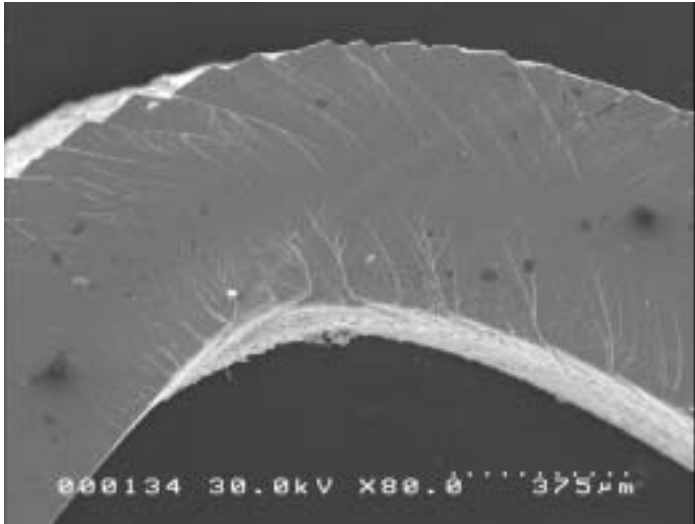
- ❑ Yielding initiates in dendrites.
- ❑ Shear bands connect precipitates.
- ❑ “Ideal” dendrite size and spacing?

Slip

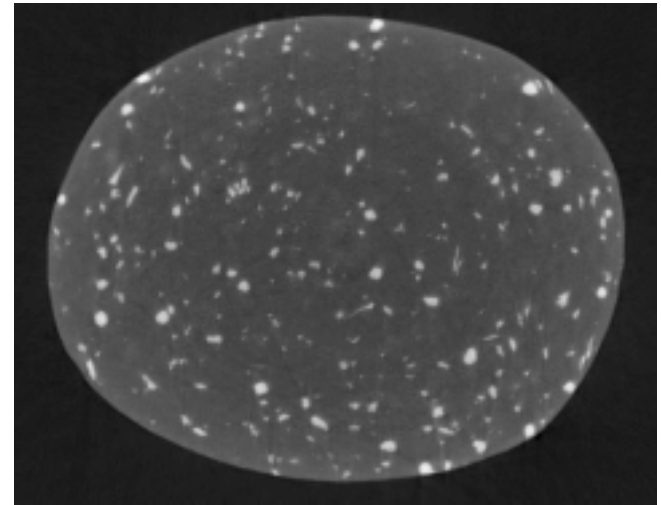


# BMG Matrix Composites: Present and Future Work

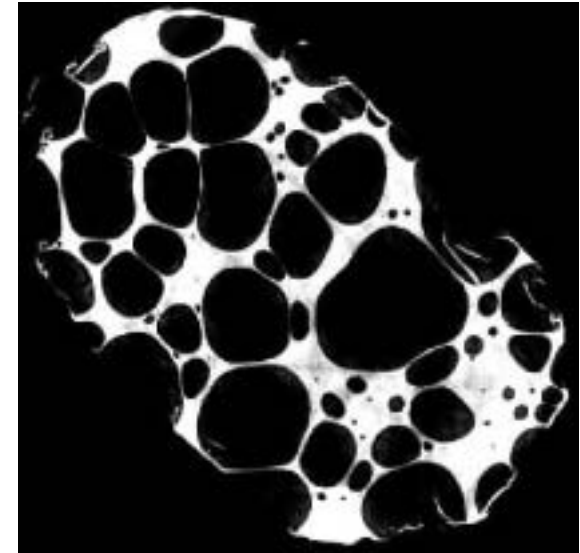
- Deformation fields in model monoliths and composites:
  - » Laser speckle interferometry (2-D)
  - » X-ray tomography (3-D)
  - » Imaging of shear bands
- 3-D FEM of deformation



BMG under bending



Tomograph of W-BMG composite

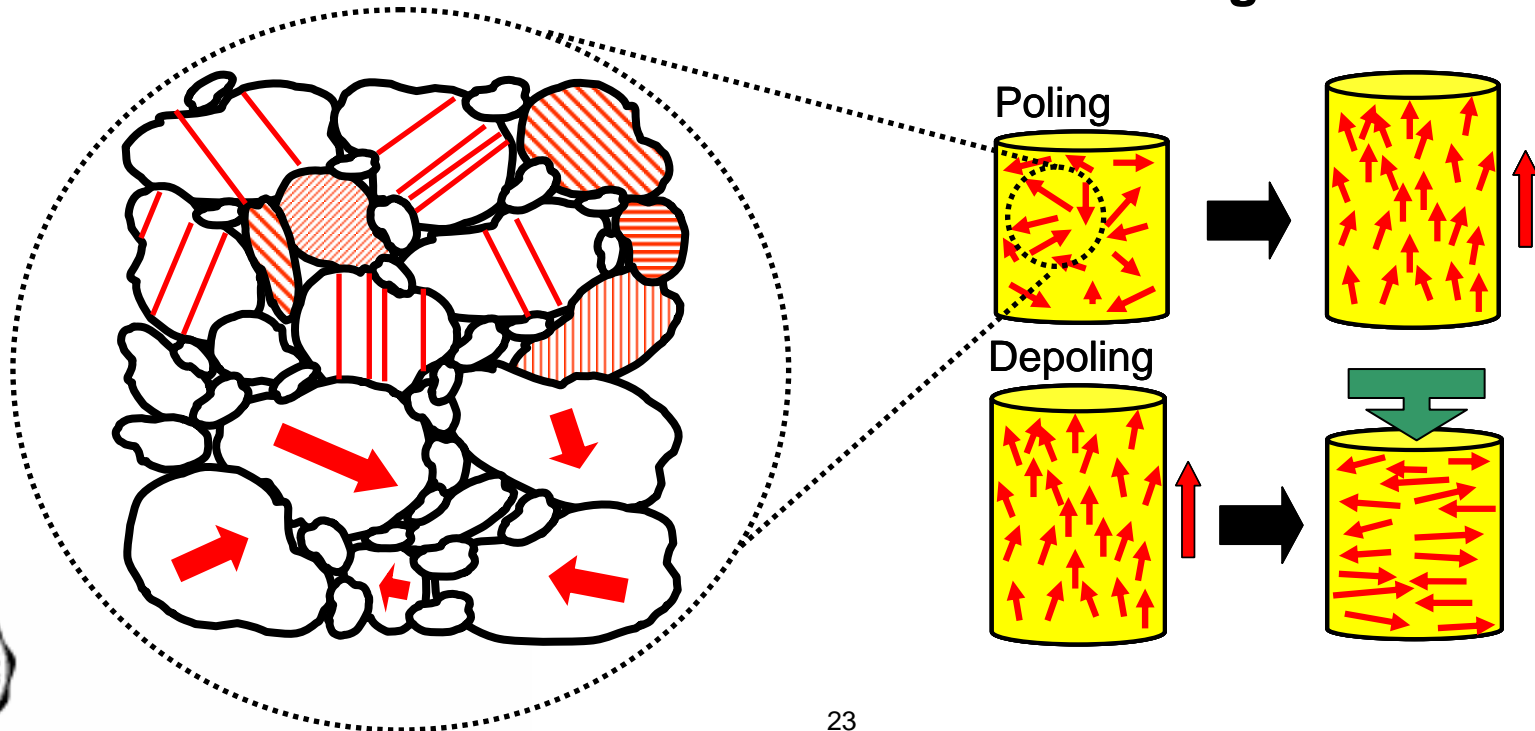


Tomograph of BMG foam



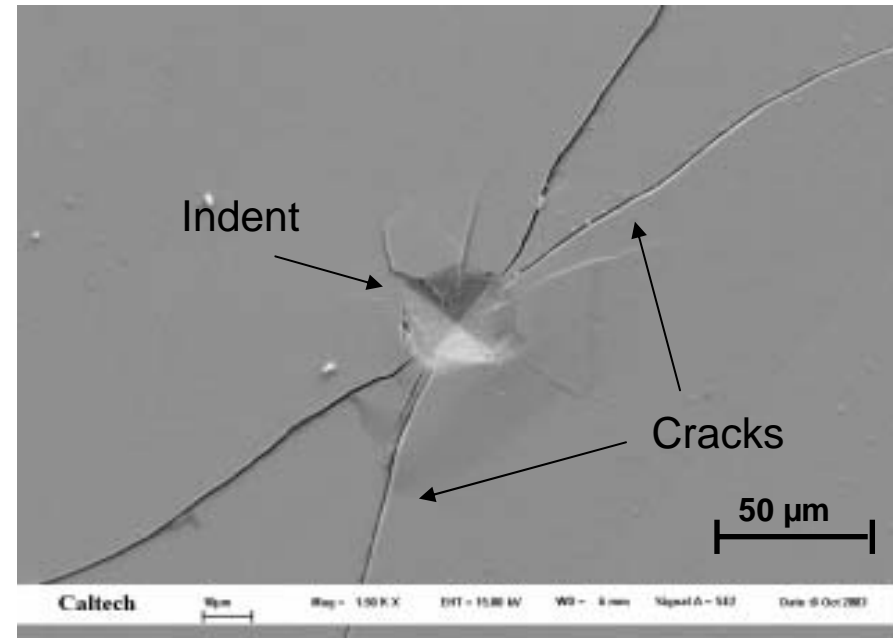
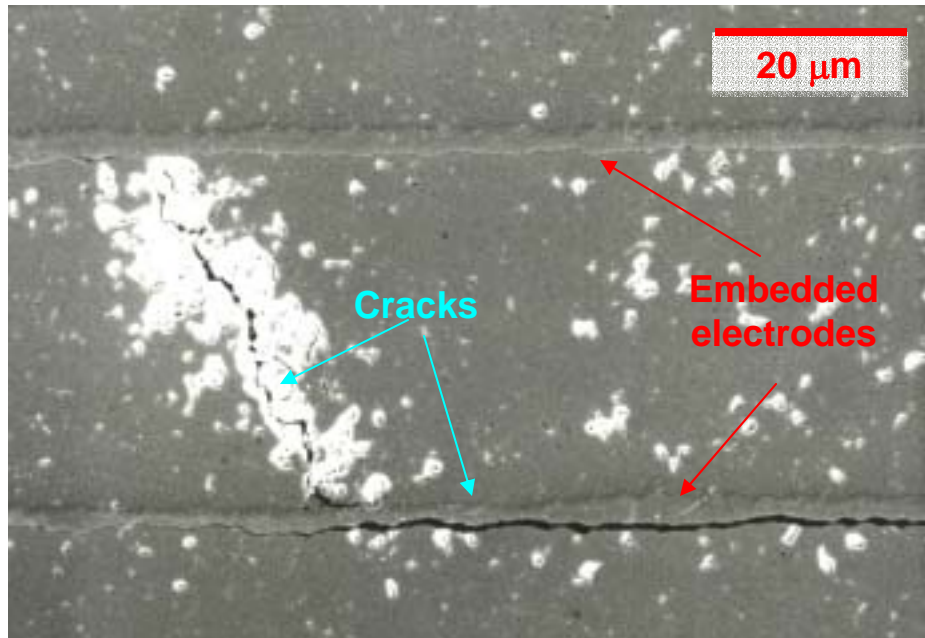
# Ferroelectrics

- Grains within a polycrystal possess randomly oriented domains.
- Electrical poling is used to align a significant number of domains and produce a technologically viable ceramic material.
- Domain motion may be constrained by grain orientation and local boundary conditions.
- What are the details of inter-domain and inter-grain interactions?





# Failure of Ferroelectrics



- Low fracture toughness ( $K_{Ic} \sim 1 \text{ MPa.m}^{1/2}$ )
- Indentation induced fracture mitigated by crystal poling direction; domain switching can impede crack propagation
- Domains interact with (and may induce) cracks
- *3-D morphology of cracks?*



# Conclusions and Future Directions

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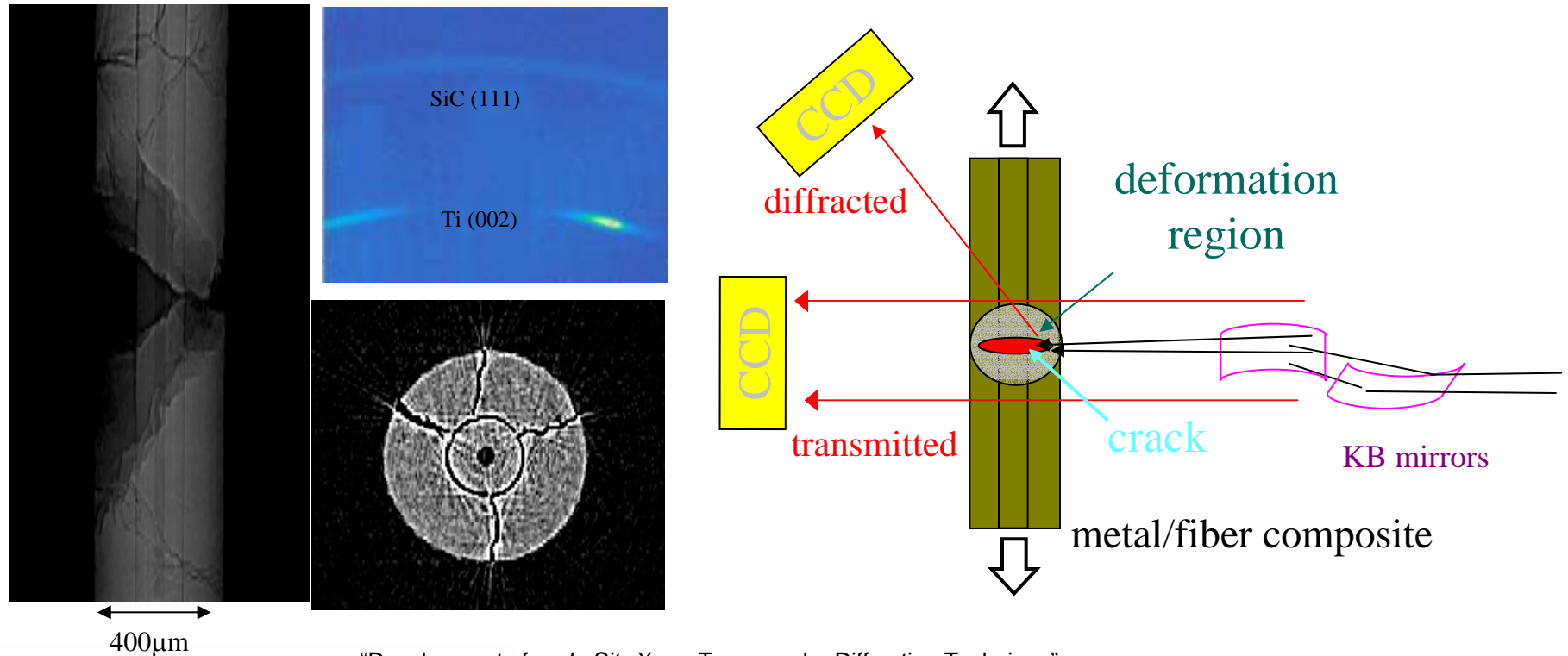
- X-ray imaging can revolutionize composites field
- Grand challenge: to monitor damage evolution in a composite in real time
- Combined diffraction and imaging capability needed
- Spatial resolution required:  $\sim 1 \mu\text{m}$  (or less)
- Real time data analysis is a necessity
- New ancillary equipment must be developed
- X-ray energy should be tunable within a wide range



# Development of *In-Situ* Tomo-Diffraction

**Goal:** develop x-ray microscopy on both real space and reciprocal space, to investigate the deformation and fracture mechanics in engineering materials.

**Strategy:** dedicated comprehensive instrument development



"Development of an *In Situ* X-ray Tomography-Diffraction Technique",  
Y. S. Chu, F. De Carlo, J. D. Almer and D. C. Mancini, SRI, 2001

"Novel X-ray Diffraction Technique for Strain Measurements Using Area Detector",  
Y. S. Chu, F. De Carlo, and D. C. Mancini, Denver X-ray Conference, 2001

